

31**730** S/081/61/000/021/035/094

B101/B147

18.

AUTHORS: Layner, D. I., Malysheva, L. A.

TITLE: The problem of corrosive destruction of copper - silicon

alloys to powder

PERIODICAL: Referativnyy zhurnal. Khimiya, no. 21, 1961, 255, abstract

21I118 (Tr. Gos. n.-i. i proyektn. in-ta po obrabotke

tsvetn. met., no. 18, 1960, 303 - 312)

TEXT: The corrosive destruction of Cu - Si alloys and the effect of temperature, atmosphere, and of the chemical composition of the alloy, upon its destruction were studied. It has been found that the alloy, if left for a long time in air at room temperature, or if annealed at 150

and 600°C in air or water vapor, crumbles and increases in weight. In some cases, the weight is increased by 40 - 50% as a result of oxidation by atmospheric oxygen. Intercrystalline cracks, which spread over to the grains of the initial Si, are formed on the polished sections by repeated annealing. In the case of long annealing, the crumbling process proceeds

Card 1/2

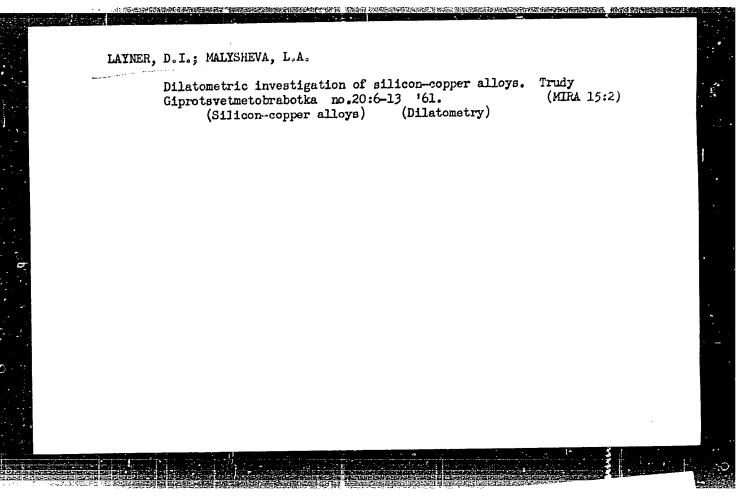
CIA-RDP86-00513R000928910003-9

31730 S/021/61/000/021/035/094 B16:/3147

The problem of corrosive ...

both along the boundaries and in the grains of Si and the intermetallic compound $\mathrm{Cu}_2\mathrm{Si}$. Oxidation is the quicker, the higher the annealing temperature, and at $600^{\circ}\mathrm{C}$ it is more intense in an $\mathrm{H}_2\mathrm{O}$ vapor atmosphere than in air. Alloys containing 40-50% Cu crumble much faster than alloys with lower or higher Cu content. It is believed that the Si contained in the alloy decomposes $\mathrm{H}_2\mathrm{O}$, whereby oxygen oxidizes Cu and Si to CuO and SiO_2 , respectively. Abstracter's note: Complete translation.

Card 2/2



s/680/61/000/020/001/013 D204/D302

Layner, D. I., Malysheva, L. A. and Sotnikova, L. 1.

Poisons of the Cu-Si alloy catalysts AUTHORS:

TITLE:

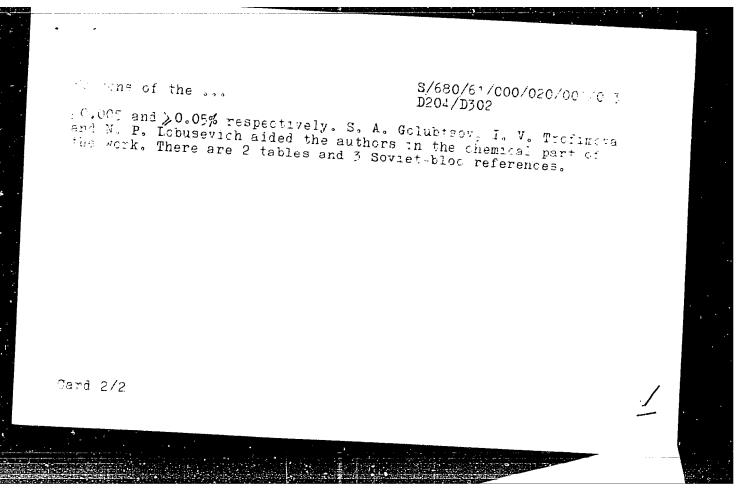
Moscow. Gosudarstvennyy nauchne issledovateľskiy proyektnyy institut obrabotki tsvetnykh metallov. Sbor nik nauchnykh trudov. no. 20, 1961, Metallovedeniye SOURCE:

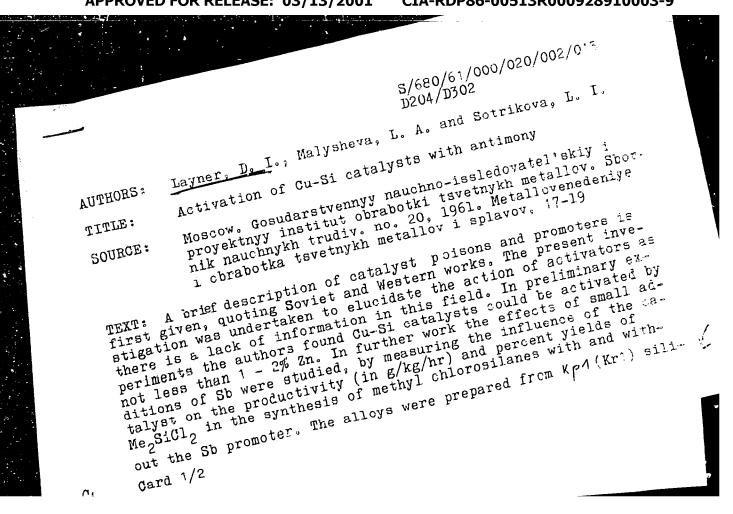
obrabotka tsvetnykh metallov i splavov, 14-16

TEXT: The authors studied the inhibiting effect of small additions of the studied the inhibiting effect of small additions are small additionally and the studied the inhibiting effect of small additionally are small additionally and the small additionally are small additionally are small additionally and the small additionally additionally are small additionally and the small additionally additional tions of Pb. Sn and Bi on the catalytic activity of 10% Cu. 90% Si alloys, by measuring the productivity (in g product/kg cata. lyst/hr) and the percentage yield of Me₂SiCl₂ in the synthesis of methyl chlorosilanes. The alloys were prepared in carbon boats from Kp1 (Kr1) silicon, MO (MO) copper, CB (SV) lead. 01 (01) tin from Kp1 (Kr1) silicon, MO (MO) copper, CB (SV) lead. 01 (01) tin from Kp1 (Kr1) silicon, MO (MO) copper, CB (SV) lead. 01 (01) tin from Kp1 (Kr1) silicon, MO (MO) copper, CB (SV) lead. 01 (01) tin from Kp1 (Kr1) silicon, MO (MO) copper, CB (SV) lead. 01 (01) tin from Kp1 (Kr1) silicon, MO (MO) copper, CB (SV) lead. 01 (01) tin from Kp1 (Kr1) silicon, MO (MO) copper, CB (SV) lead. 01 (01) tin from Kp1 (Kr1) silicon, MO (MO) copper, CB (SV) lead. 01 (01) tin from Kp1 (Kr1) silicon, MO (MO) copper, CB (SV) lead. 01 (01) tin from Kp1 (Kr1) silicon, MO (MO) copper, CB (SV) lead. 01 (01) tin from Kp1 (Kr1) silicon, MO (MO) copper, CB (SV) lead. 01 (01) tin from Kp1 (Kr1) silicon, MO (MO) copper, CB (SV) lead. 01 (01) tin from Kp1 (Kr1) silicon, MO (MO) copper, CB (SV) lead. 01 (01) tin from Kp1 (Kr1) silicon, MO (MO) copper, CB (SV) lead. 01 (01) tin from Kp1 (Kr1) silicon, MO (MO) copper, CB (SV) lead. 01 (01) tin from Kp1 (Kr1) silicon, MO (MO) copper, CB (SV) lead. 01 (01) tin from Kp1 (Kr1) silicon, MO (MO) copper, CB (SV) lead. 01 (01) tin from Kp1 (Kr1) silicon, MO (MO) copper, CB (SV) lead. 01 (01) tin from Kp1 (Kr1) silicon, MO (MO) copper, CB (SV) lead. 01 (01) tin from Kp1 (Kr1) silicon, MO (MO) copper, CB (SV) lead. 01 (01) silicon, MO (MO) copper, CB (SV) lead. 01 (01) silicon, MO (MO) copper, CB (SV) lead. 01 (01) silicon, MO (MO) copper, CB (SV) lead. 01 (01) silicon, MO (MO) copper, CB (SV) lead. 01 (01) silicon, MO (MO) copper, CB (SV) lead. 01 (01) silicon, MO (MO) copper, CB (SV) lead. 01 (01) silicon, MO (MO) copper, CB (SV) lead. 01 (01) silicon, MO (MO) copper, CB (SV) lead. 01 (01) silicon, MO (MO) copper, MO (MO) co Ph. Bi and Sn poisoned the catalyst when in quantities > 0.003.

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CIA-RDP86-00513R000928910003-9" APPROVED FOR RELEASE: 03/13/2001





S/680/61/000/020/003/013 D205/D302

AUTHORS: Layner, D. I. and Krupnikova-Perlina, Ye. I.

TITLE: On the oxidation mechanism of copper at high tempera-

tures

SOURCE: Moscow, Gosudarstvennyy nauchno-issledovatel'skiy i pro-

yektnyy institut obrabotki tsvetnykh metallov. Sbornik nauchnykh trudov. no. 20, 1961. Metallovedeniye i obra-

botka tsvetnykh metallov i splavov, 20-27

TEXT: This work was performed to verify the opinions of M. T. Mishchenko and R. R. Gorain of the Livov Polytechnic Institute. 99.996% pure Cu samples 0.75 x 24 x 100 mm were subjected to oxidation at 1000°C over 5, 10, 15, 20 and 30 hours, by air at atmospheric pressure. Examination of the samples after various times of oxidation leads to the following picture of the mechanism: The exidation takes place by diffusion of Cu across the product Cu_2O layer. It can be assumed that the formation of elongated Cu_2O

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On the oxidation mechanism ...

S/680/61/000/020/003/0:3 D205/D302

crystals is the outcome of this diffusion. Grain growth, cracks and imperfections develop across the grain boundaries and permit the access of oxygen to the remaining metal. Small polyhedral Cu₂O crystals are formed. At a given moment, this second mechanism may become prevalent. There are 6 figures and 17 references: 13 Sciviet-bloc and 4 non-Soviet-bloc. The reference to the English language publication reads as follows: Moore and Selicson, J. of Chem. Phys., v. 19, mo. 12 (1951).

Card 2/2

35041 s/680/61/000/020/004/013 D_{205}/D_{302}

AUTHORS:

Layner, D. I. and Tsypin, M. I.

TITLE:

Some structural features of scale formation on titanium

SOURCE:

Moscow. Gosudarstvennyy nauchno-issledovatel'skiy i proyektnyy institut obrabotki tsvetnykh metallov. Sbornik nauchnykh trudov. no. 20, 1961. Metallovedeniye i obra-botka tsvetnakh metallov i splavov. 28-41

TEXT: The structural characteristics of the scale are the basis on which one's notions on the diffusion mechanism in the oxidation process are formed. Parting from this point of view, investigations were carried out by the same authors on the oxidation of titanium (Ref. 1: Sb. nauchnykh trudov instituta "Giprotsvetmetcbrabetka", no. 20, Metallurgizdat, 1961, this collection, pp. 42-64). The present article is connected with the methods of data. sent article is connected with the methods of determining the scale-formation features. The methods of calculating the interplanar distances of titanium dioxide was based on Ormont's work (Ref. 6: Struktury neorganicheskikh veshchestv (Structures of Inorganic

Card 1/2

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S/680/61/000/020/004/013 D205/D302

Some structural features ...

Substances). Gostekhteoretizdat, 1950) and of the intensivities of V. K. Vainshteyn's book (Ref. 7: Strukturnaya elektronografiya (Structural Electronografiy), AS USSR, 1956). The preferential contentation of the rutile lattice during various regimes of exidation is discussed in relation to the electronographic images. It is argued that structure of the titanium dioxide scale explains the diffusion mechanism. Some of the conclusions pertaining to this article are given in Ref. 1 (Op. cit.) which is the following paper in this collection. There are 16 figures, 3 tables and 10 references: 7 Sovie -bloc and 3 non-Soviet-bloc.

Card 2/2

35042 \$/680/61/000/020/005/013 D205/D302

18.1285 AUTHORS:

Layner, D. I. and Tsypin, M. I.

TITLE:

Investigating the structure of titanium scale during

its formation

SOURCE:

Moscow. Gosudarstvenny nauchno-issledovatel skiy i proyektnyy institut obrabotki tsvetnykh metallov. Sbornik nauchnykh trudov. no. 20, 1961. Metallovedeniye i obra-

botka tsvetnykh metallov i splavov. 42-64

TEXT: The oxidation of titanium is of practical and theoretical interest. On the practical side, it is connected with the problem of corrosion resistance and with some technological properties, for example antifrictional, of oxide layers. On the theoretical side, many problems still remain unsolved. One of these is the relative roles of the titanium and oxygen ions in the diffusional process, the other is the cubic relationship of the rate of oxidation between 300 and 600°C. The oxidized films were separated from the metal by the use of the etching mixture: 35 ml HCl_{ϕ}^{2} 0.: - : 0

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Investigating the structure ...

g NaF, 65 ml H₂O. After separation, the floating oxide films were removed with a glass spatula, washed twice in distilled water, dried and placed in the electronograph. Good electronograms were obtained which showed that only one phase, that of TiO₂, was present a making rooms of the film was determined from the electronogram

sent. Thickness of the film was determined from the electronograms and plotted vs. time of oxidation for various oxidation temperatures in the 350 - 750°C range. Films obtained at higher temperatures were examined by X-rays and microscopically, owing to their thickness which made them unsuitable for electronographic investigation. In addition, samples of sheet titanium were oxidized in steam at 450, 650 and 850°C. The oxidation of Ti in the 300 - 600°C range followed the cubic relationship and generally confirmed the mechanism proposed by Cofstad et al. (Ref. 1: Acta Chem. Scand. 12, 239, 1958), based on the diffusion of oxygen into the metal from the oxide. Above 600°C a prevailing orientation begins to appear on the external surface of the oxide, pointing to this surface as the locus of the oxide layer formation. Simultaneously the parabolic oxidation becomes prevalent and the diffusion of Ti ions through

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LAYNER, D.I.; TIKHONOV, B.S.; KRUPNIKOVA-PERLINA, Ye.I.; AGAFONOVA, A.V.

Investigations in the field of improving service characteristics of zinc for printing purposes. Trudy Giprotsvetmetohrabotka no.20:97-103 '61. (MIRA 15:2)

(Zinc-Metallurgy)

	TRA	NSLATION	S/180/61/000/001/014/015 E073/E535
	AUTHORS: Layner,	D. I. and Tsypin	, M. I. (Moscow)
0	TITLE: On the 0 300 to 6	xidation of Tita	$\frac{1}{2}$
	PERIODICAL: Izvestiy nauk, Me	a Akademii nauk tallurgiya i top	SSSR, Otdeleniye tekhnicheskik livo, 1961, No.1, pp.146-147
25 25	nauk, Metallurgiya i toplivo, 1961, No.1, pp.146-147 TEXT: Usually the "weight increment-time" relations of oxided metallic specimens as a function of temperature are straight lines, parabolas or logarithmic curves. Existing oxidation theories by Wagner, Dankov, Mott and others propose oxidation mechanisms that lead to these relations. However, P. Cofstad, K. Hauffe and H. Kjöllesdal (Ref.1) have established that in oxidation of titanium in the temperature range 300 to 600°C the dependence of the weight increase on time is represented by a cubic parabola, a feature which has not been observed in the oxidation of other metals. Titanium can absorb up to 30 at.% of oxygen which dissolves in the metal and Cofstad et al. proposed a mechanism of oxidation which explains the experimentally determined time dependence of the weight increase by diffusion of oxygen in metallic titanium through		
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(10) (1)	•	•	

On the Oxidation of Titanium S/180/61/000/001/014/015 E073/E535

an oxide film of constant thickness. Detection of the existence of such films would be a decisive factor in judging whether this mechanism does really take place. For this purpose the authors used a very simple and sufficiently accurate method of determining the thickness of oxide films from their interference colours. Spectrum analysis of arc smelted titanium sheet specimens showed the presence of about 0.04% Si and other metallic admixtures in quantities below 0.01% each (a total of 12 elements were determined). After vacuum annealing (10⁻¹ mm Hg) for 30 mins at 800°C the specimens were cleaned with alcohol and oxided in air. The correspondence of the well known coloration of oxide films on the titanium for the temperature range 350 to 700°C withinterference phenomena is confirmed by the table, which indicates the sequence of alternating the colours of the Newton ring and the colours of the wedge-shaped layer of oxides at the surface of the titanium strip, one end of which was in air, whilst the second end was in a furnace at 800°C. The film thickness 6 is determined by means of the expression

 $\delta = k - \frac{\lambda}{4n}$

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On the Oxidation of Titanium ...

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where k = 1, 2, 3... is the order of reflection, λ is the wavelength of the particular colour which is complementary to the colour of the film, n - refractory index of the film. equation the error in the visual determination of $\ \lambda$ is about In this 1.5 to 2% (10 to 15 μ at wavelengths of 400 to 700 $\mu)$. The main error is introduced in the averaging of the refraction coefficient for optically uniaxial crystals of TiO2-rutile, of which the oxide layers are formed (see earlier work of the authors, Refs. 3 and 4), as well as by not taking into consideration dispersion. average value $n_{av} = 2.8$ if the maximum scatter is 2.6 to 3. Thus, the total random error in the thickness of the film can be evaluated at 10%. Furthermore, a systematic error is introduced due to the absence of accurate data on the conditions of reflections at the boundary oxide-metal; however, this effect is of importance only for the absolute film thicknesses but does not manifest itself on the characteristic form of the curves "film thickness-time". The calculated values of the film thickness were qualitatively confirmed by the change in the character detected in recent electron diffraction investigations of the authors (Ref. 4). Card 3/6

On the Oxidation of Titanium ... S/180/61/000/001/014/015 E073/E535

The obtained film thickness values given in the graphs (top graph a - for the temperature range 350 to 600°C; bottom graph b - for the range 550 to 700°C; b, & vs. \(\tau\), min) show clearly that if titanium is oxided in the temperature range 350 to 550°C the forming oxide films will reach a practically constant thickness and, consequently, the results confirm the correctness of the mechanism for oxidation of titanium in the intermediate temperature range (about 300 to 600°C) proposed by Cofstad et al. There are 1 figure, 1 table and 4 references: 3 Soviet (1 a translation from English) and 1 non-Soviet.

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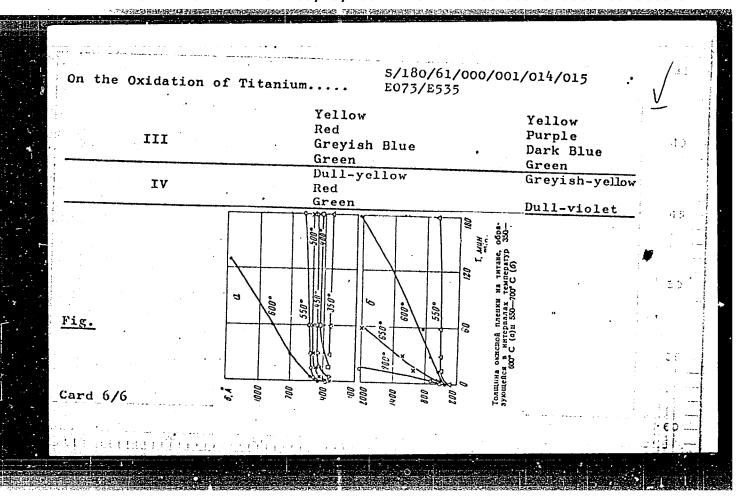
(Note: this is a complete translation)

SUBMITTED: January 28, 1960

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S/180/61/000/001/014/015 Sequence of Basic Colours of the Wedge-shaped Oxide Layer on Titanium			
Order of reflection	Colour of the wedge- shaped air gap (Newton ring) in transmitted light	Colour of oxides on titanium in reflected light	;
I	Yellow Red to Violet Dark Blue	Yellow Brown Violet Dark Blue	i
II	Green Yellow Red *	Pole Blue Greenish Yellow Purple	
ard 5/6	Dark Blue Green	Violet Dark Blue Green	3.2
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"APPROVED FOR RELEASE: 03/13/2001 CIA-RDP86-00513R000928910003-9



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S/136/61/000/002/003/006 E073/E335

AUTHORS: Layner, D. I. and Golubtsova, L.M.

TITLE: Means of Improving the Thermo-electric Properties

of the Semiconductor Alloy Zinc-antimony

PERIODICAL: Tsvetnyye metally, 1961, No. 2, pp. 69 - 74

TEXT: In the first approximation the efficiency of a material for producing the mo-electricity can be estimated on the basis of the product α where α is the thermo-electric coefficient, $\mu V/C_1$ or is the specific electrical

conductivity ohm cm cm Classical materials for producing thermo-electricity are: intermetallic ZnSb compounds and electron lead sulphide. ZnSb compounds form by a peritectic reaction at 545 °C; thereby the concentration of the components is 35 wt.% Zn and 65 wt.% Sb. Microscopic analysis indicates that for a Zn content below 35 wt.% the alloy contains Sb abd SbZn which may be present as a eutectic and as larger crystals. Alloys containing Zn in quantities of 35-45 wt.% always contain a mixture of crystals of the two intermetallic Card 1/8

Means of Improving

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compounds $SbZn(\alpha)$ and $Sb_3Zn_{\underline{4}}(\beta)$. The compound SbZn is a hole-type semiconductor (this compound was investigated by Ye.D. Devyatkova, Yu.I. Maslakovets and I.V. Mochan of the AS USSR). In determining the electrical conductivity various investigators used specimens of different sizes. On small specimens I.V. Mochan (Ref. 1) determined the values $\sigma = 4 \times 10^{-3}$ and on larger specimens Ye.D. Devyatkova and Yu.I. Maslakovets (Ref. 2) obtained the value equalling 80 chm⁻¹cm⁻¹. The width of the barrier zone equals 0.7 - 0.8 eV (Ref. 1); the thermo-e.m.f. of the compound equals 180 - 220 $\mu V/deg$ and even the smallest shift in the ratio of the components reduces it considerably. The authors carried out experiments with a view to improving the thermoelectric properties of ZnSb compounds by introducing admixtures of various elements. A total of 150 ZnSb compounds were tested which contained admixtures of various elements, both separate and combined: Cu; Ag; Bi; Si; Ge; Sn; Pb; In; Te; Al; Cd and Fe. The melting was in a high-fre The melting was in a high-frequency

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Means of Improving

induction furnace inside graphite crucibles. The basic components were introduced into the charge designed to produce 500 g of the alloy with an excess of zinc of 0.4% compared with the stoichiometric composition. Melting was carried out at 750-800 °C with careful mixing during the process. Teeming was into metallic moulds at a temperature of the order of 700 The sequence of feeding the charge was as follows: a) Zn, admixtures, Sb for alloys with additions of lowmelting point metals; b) charging Zn and Sb simultaneously and introducing the admixtures into the molten metal; this was done in the case of high melting-point admixtures. From the alloys 8 x 10 x 15 mm specimens were pressed (4 tons/cm²) at 400 °C. The electric conductivity or was measured by a compensation method, using probes for tapping-off the voltage. The thermo-e.m.f. α was also measured by a compensation method relative to lead with a temperature difference of 18-20 °C. It was found that a great increase in the electric conductivity can be achieved by using additions of Cu, Ag and Ge. The obtained results for Cu are graphed in Fig. 1 Card 3/8

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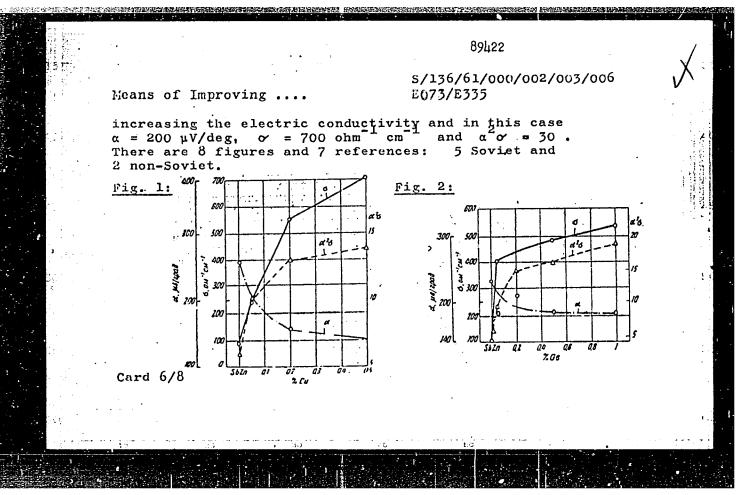
Means of Improving

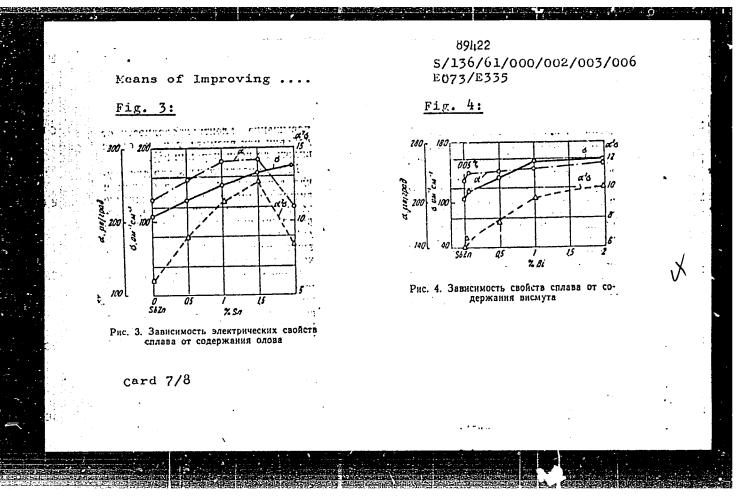
(α , $\mu V/\text{deg}$ and σ , ohm⁻¹cm⁻¹ as functions of the Cu content, %) for additions of Cu; in Fig. 2 - for additions of Ge (same notation); in Fig. 3 - for additions of Sn; in Fig. 4 for additions of Bi and in Fig. 5 - for additions of In. In contrast to results of Devyatkova and Maslakovets (Ref. 2), according to which the "hole" concentration reaches a maximum, for a content of 0.5% Sn the authors of this paper found a continuous increase in the electric conductivity with increasing Bi and Sn contents up to concentrations of 2%. Microstructural and X-ray-structural analyses did not reveal the presence of any new phase differing from SbZn and Sb3Zn4. investigated alloys showed a qualitatively similar picture in the angular range $10\text{--}40^{\circ}$. However, certain changes in the intensity of the lines corresponding to SbZn were observed. Obviously, the change in properties is due to complicated structural processes in the crystal lattice, which require application of finer methods of analysis. Nomograms of the change in the properties of materials alloyed with elements of various groups of the periodic system indicate that the best results Card 4/8

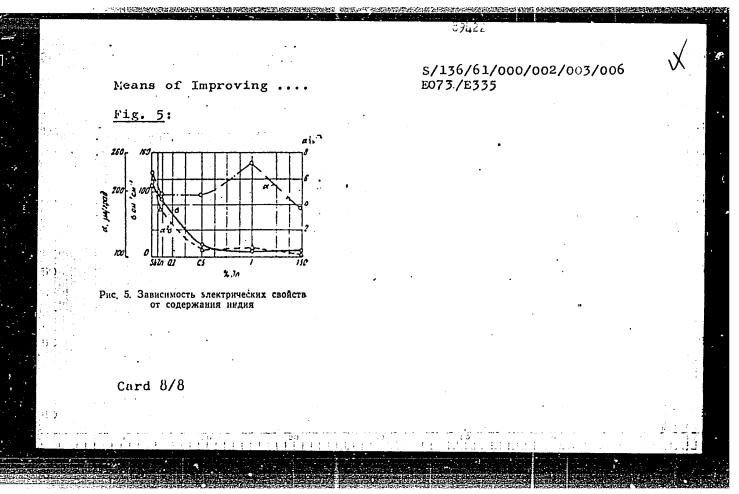
Means of Improving

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can be achieved if Ag, Cu or Ge are used. A considerable increase in the thermo-e.m.f. is achieved by additions of Sn and Si and a less intensive increase is achieved by the addition of Pb and Bi. An increase in the thermo-e.m.f. in the case of an alloy with Cd only or with Si only is accompanied by a sharp drop in the electric conductivity and is not of practical interest. For practical purposes, the best properties are achieved with additions of Sn, Pb, Bi in which the high thermo-e.m.f. are combined with a sufficiently high electric conductivity. Still better results can be obtained by alloying with several elements since, separately, none of these is able to ensure satisfactory properties. Good results were achieved with an alloy containing (% of the weight of the SbZn compound) 35 Zn, 65 Sb, 2 Bi and 1.5 Sn. By means of this alloy the following results were achieved: $\alpha = 200-300 \,\mu\text{V/deg}$ for σ not less than 200 ohm⁻¹ cm⁻¹ and $\alpha^2 \sigma = 18-20$. can be used successfully in the anode blocks of thermal batteries; 0.5% Ag is introduced into the heating blocks for Card 5/8





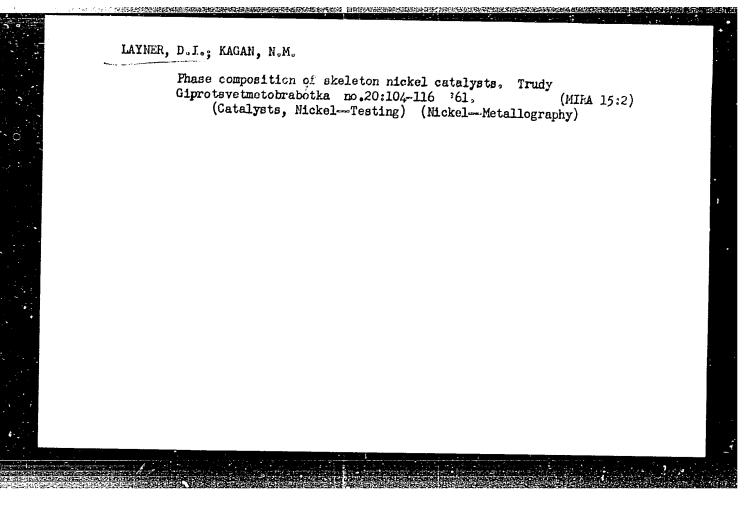


LAYNER, D.I.; BAZHENOVA, L.A.; AGAFONOVA, A.V., Prinimali uchastiye:
PAKHOMOVA, Ye.F., inzh.; KORSUNSKAYA, K.N., inzh.

Effect of various additions on the modification and recrystallization temperature of zinc. Trudy Giprotsvetmetobrabotka no.20:81-96

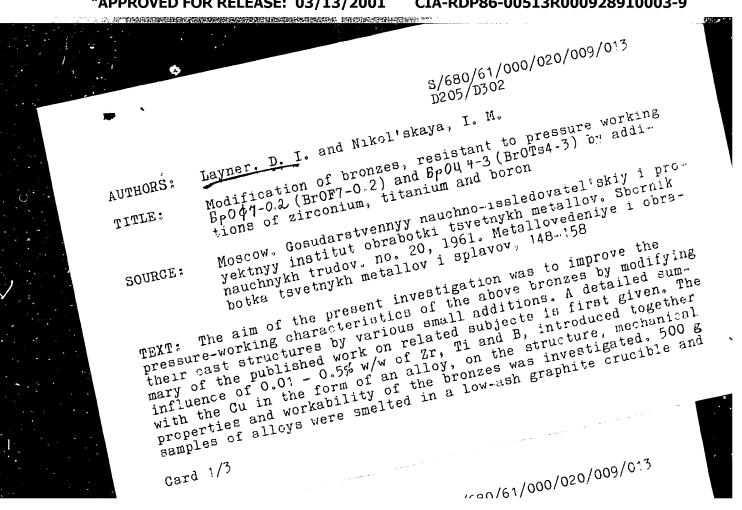
'61. (MIRA 15:2)

(Zinc-Metallurgy) (Crystallization)



LAYNER, D.I.; DRUPNIKOVA-PERLINA, Ye.I.; BAY, A.S.

Electron microscopy for the determination of metal texture. Trudy Giprotsvetmetobrabotka no.20:142-147 '61. (MIRA 15:2) (Metallography) (Electron microscopy)



3/680/61/000/020/002/013 D204/D302

AUTHORS: Layner, D. I., Malysheva, L. A. and Sotrikova, L. I.

TITLE: Activation of Cu-Si catalysts with antimony

SOURCE: Moscow, Gosudarstvennyy nauchno-issledovatel'skiy i proyektnyy institut obrabotki tsvetnykh metallov. Stornik nauchnykh trudiv. no. 20, 1961. Metallovenedeniye

i obrabotka tsvetnykh metallov i splavov, 17-19

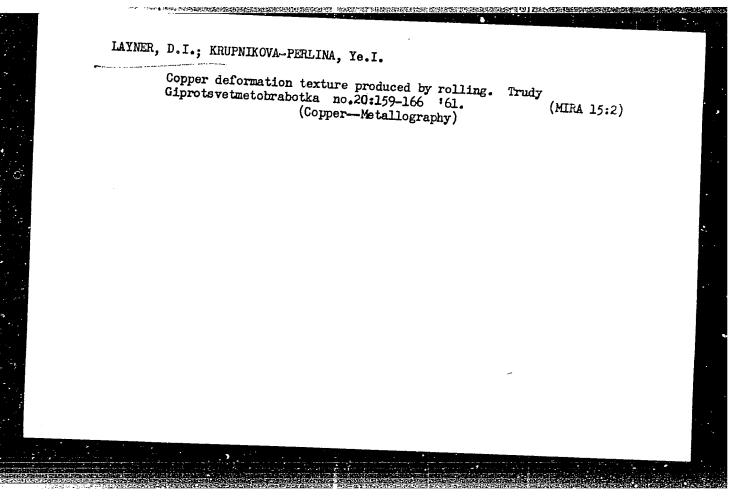
TEXT: A brief description of catalyst poisons and promoters is first given, quoting Soviet and Western works. The present investigation was undertaken to elucidate the action of activators as there is a lack of information in this field. In preliminary experiments the authors found Cu-Si catalysts could be activated by not less than 1 - 2% Zn. In further work the effects of small additions of Sb were studied, by measuring the influence of the catalyst on the productivity (in g/kg/hr) and percent yields of MegSiCl2 in the synthesis of methyl chlorosilanes with and without the Sb promoter. The alloys were prepared from Kp4 (Kr1) siling Card 1/2

Activation of Cu-Si ...

\$/680/61/000/020/002/C13 D204/D302

con, electrolytic MO (MO) copper and CyA (Su!) antimony, by fusion in graphite boats at 1550-1600°C, without fluxes, and contained 5.10 or 20% Cu and 0.005, 0.01, 0.02 or 0.05% Sh. Antimony was increasing the yield ~1.5 times, and as a catalyst poison in the ingots produced by (1) the experimental 'Giprotsvetmetobrabetka' tsov, I. V. Trofimova and N. P. Lobusevich aided the authors in the chemical part of the work. There are 1 figure, 1 table and 4 rethe English-language publications read as follows: T. Griffiths, and A. Reinan, Ind. Eng. Chem., 20, 1105, 1928.

Card 2/2



S/126/61/012/003/010/021 E193/E135

CONTROL OF THE STATE OF THE STA

AUTHORS: Layner, D.I., and Slesareva, Ye.N.

TITLE: The effect of tin on oxydation of titanium

PERIODICAL: Fizika metallov i metallovedeniye, vol.12, no.3, 1961, 395-402

Although tin is added to titanium-base alloys to TEXT: stabilise the a-phase, to ensure good weldability, and to improve the workability of titanium-base alloys with a high aluminium content, little is known about the effect of this element on oxydation of titanium. A.E. Jenkins (Ref.l: J. Inst. Metals, 1955, Vol.84, No.1, 1) who had studied the 11.4% Sn-Ti alloy, found that in the presence of Sn the rate of oxydation of titanium at high temperatures (starting from 850 °C) rapidly increased. He postulated that the ability of oxygen to diffuse through the scale and through the metal under the scale is the governing factor in the process studied. On the other hand, the results obtained by some Soviet workers indicated that diffusion of titanium plays a predominant part in the formation of titanium scale. The object of the present investigation was to check the Card 1/6 -

The effect of tin on oxydation of ... S/126/61/012/003/010/021 E193/E135

results obtained by Jenkins. The experimental work was carried out on binary alloys containing 0.02, 1.65 or 9.0 wt.% Sn, melted in a vacuum-arc furnace, forged, rolled, and annealed in air. The oxydation tests were conducted on polished specimens $10 \times 10 \times 15$ mm, the increase in weight being used as the measure of the degree of oxydation. The results are reproduced in Fig.1, where the increase in weight (mg/cm2) is plotted against time (au, hours) at temperatures indicated by each curve. The experimental points denoted by circles, dots, crosses and triangles relate, respectively, to pure titanium and to titanium alloys containing 0.02, 1.65 and 9.0% Sn. It will be seen that the rate of oxydation of titanium, practically unaffected by 0.02 and 1.65% Sn additions, increased more than fivefold (at 1000 °C) in the presence of 9% Sn. This difference was also reflected in the results of X-ray diffraction analysis of the scale. Scale formed on the former two alloys consisted almost exclusively of TiO2, whereas that formed at 1000 °C on the 9% Sn alloy consisted of the following four layers: textured TiO2; TiO2 with no texture; TiO; Sn at the TiO/alloy interface. Another effect of the Card 2/8,-

The effect of tin on oxydation of.. $\frac{S}{126}/61/012/003/010/021$

presence of a large proportion of tin in the alloy was revealed by microhardness measurements. It was found that microhardness of the alloy layer saturated with oxygen (that is the layer adjacent to the scale) varied depending on the Sn content, being 945, 675, and 300 kg/mm² in the 0.02, 1.65 and 9% Sn-Ti alloys, respectively. The conclusions reached by the present authors can be summarised as follows. In the presence of small quantites of tin the mechanism of oxydation of the Ti-Sn alloys is the same as that for pure titanium: when the tin content is high the process of exydation also begins by the formation of TiO2, but at the same time tin which is surface-active in respect to titanium diffuses towards the surface metal layers. As a result, the concentration of Sn in the surface layer may increase to such an extent that localised melting of the alloy takes place. Since diffusion through a liquid face proceeds at a rate considerably faster than through the solid metal, this effect would explain the rapid increase of the oxydation rate in the Sn-rich titanium alloys. It is also possible that the outward diffusion of titanium causes the formation of vacancies in the interior of the specimen, whereby

The effect of tin on oxydation of ... S/126/61/012/003/010/021 E193/E135

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diffusion of titanium is facilitated. This, combined with the increase in the rate of diffusion due to the presence of a liquid phase, leads to the formation of an internal TiO layer at the Sn/TiO2 interface, which means that, contrary to the conclusions of Jenkins, counter-current diffusion is taking place. indicated not only by the presence of an Sn layer at the metal/ scale interface, but also by the absence of oxygen-saturated metal layer under the scale, formed on the Sn-rich alloys. The latter effect indicates that, in the presence of a liquid, Sn-rich phase, diffusion of titanium increases to such an extent that practically all oxygen diffusing through the scale is taken up by titanium diffusing outward from the alloy. Thus, the catastrophic rate of oxydation of the 9% Sn-Ti alloy at high (> 700 °C) temperatures must be attributed to the outward diffusion of titanium which can take place owing to the formation of a layer of a liquid, Sn-rich phase. There are 4 figures, 2 tables and 7 references: 3 Sovietbloc and 4 non-Soviet-bloc. The English language references read: Ref. 1: as quoted in the text above. Ref. 6: M. Hansen. Constitution of Binary Alloys. 1958, p. 1212. Card 4, B -

The effect of tin on oxydation of ... S/126/61/012/003/010/021 E193/E135

ASSOCIATION: Giprotsvetmetobrabotka pri VSNKh (Giprotsvetmetobrabotka at VSNXh)

SUBMITTED: January 27, 1961

Card 5/6 -

L 13063-63 BDS/EWP(q)/EWT(m) AFFTC/ASD ACCESSION NR: AT3003009 S/2927/62/000/000/0228/0235 AUTHOR: Meskin, S. S.; Layner, D. I.; Kogan, L. M.; Trushina, V. Ye.; Libov, L. D. TITLE: V Titanium rectifiers [Report of the All-Union Conference on Semiconductor Devices neld in Tashkent from 2 to 7 October 1961] SOURCE: Elektronno-dy*rochny*ye perekhody* v poluprovodnikakh. Tashkent, Izd-vo AN UZSSR, 1962, 228-235 TOPIC TAGS: titanium rectifier ABSTRACT: Construction, physical phenomena, and results of testing of titanium rectifiers (manufactured in USSR since 1959) are reported. Electrophysical data of the source material, rutile, is given. Current-voltage characteristics (for 20, 150, and 250C), reverse-current-density, forward-voltage drop, cutoff voltage, differential resistance, capacitance, and barrier-layer width as functions of temperature (20-250C) are presented. Also resistance-voltage curves are given for the above 3 temperatures and within -5 +2 v. The following data that can be considered as ratings are supplied: operating temperature range -60 +250C; working voltage per element 11-25 v amplitude; reverse-current density 4, 6, and 8 ma per sq cm at -60, +20C, 150C, and 200-250C respectively; forward-current density Gard 1/2

100-200 ma per sq c			7		
= rormoras, and 5 f	.00-200 ma per sq cm; life 5,000 hrs or more at 200. Orig. art. has: 9 figur				
ASSOCIATION: Akade		Sciences ESER); Akademiya n Shkentskiy gosuderstvenny*y			
SUBMITTED: 00	DATE ACQ: 15May63	ENCL: 00			
SUB CODE: 00	NO REF SOV: OOL	OTHER: 007			
Card 2/2					

33166

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5/136/62/000/002/003/004 E021/E135

AUTHORS:

Layner, D.I., and Slesareva. Ye.N.

TITLE :

The influence of some alloying additions on the

oxidation of titanium

PERIODICAL: Tsvetnyye metally, no.2, 1962, 70-76

TEXT : The influence of 0-15% Al, 0-9% Zr, 0-17.7% W, 0-8.7% Nb, 0-17.7% Ta, 0-3.2% Si, 0-8% Mo, 0-6% V, 0-6% Mn and 0-9% Sn was investigated. The change in weight during exidation was determined and phase analysis of the scale and metallic layer under the scale was carried out. Microhardness determinations to find the depth of penetration of oxygen were also made. Results showed that W, Ta, Nb, Si and Mo lower the rate of increase in weight; Zr, Mn and Al have little effect on the weight increase; and V and Sn increase the rate of change in weight. Fig.1 shows the increase in weight $(mg/cm^2 10^{-4})$ of alloys containing W, Ta, Nb, Si and Mo during oxidation against time (hours). After 7 hours at 900 °C alloys containing Si and W form a thicker scale (0.035-0.040 mm) than alloys containing Ta or Nb (0.01-0.15 mm). Unalloyed Ti has a scale 0.1 mm thick.

33166

The influence of some alloying ... S/136/62/000/002/(003/004) E021/E135

Oxygen penetrated the alloy containing Si to a depth of 0.08 mm, the alloy containing Ta to 0.15 mm, and the alloy containing Nb to a depth of 0.30 mm. The alloy containing W had oxygen penetration to a depth of 0.65 mm. There is a layer of fine grained WO₃ underneath the layer of coarse grained rutile. Fig. 5 shows the increase in weight during oxidation against time for alloys containing Al and Zr. The scale on the alloy containing Al consists only of rutile. There are additional lines on the X-ray photographs which do not correspond to either Al or Al₂0₃. Fig. 7 shows the increase in weight of alloys containing V, Sn and Mn against time. Phase analysis of the scale on the alloy containing tin consisted of TiO₂ on the surface, TiO under the rutile and metallic tin at the metal-scale interface. The phase analysis of the alloy containing V showed only rutile. There are 8 figures, 1 table and 12 references: 4 Soviet-bloc and 8 non-Soviet-bloc. The four most recent English language references read as follows: Ref.1: Steel, v.143, no.17, 1958, 46.

Card 2/ 3

The influence of some alloying ... S/136/62/000/002/003/004 E021/E135

Ref.3: H.W. Maynor, R.E. Swift, Corrosion, no.6, 1956, 49. Ref.6: H. Margolin, Metal Progress, v.71, no.2, 1957, 86. Ref.9: F.W. Fink, R.S. Peoples. Titanium Metallurgical Laboratory, Battele Memorial Institute TML, 1956, I, 29, pp.60, 30, ICI, v.2, no.3, 1956, 95.

45226 S/806/62/000/003/010/018 AUTHORS: Layner, D.I., Tsypin, M.I. On the first stages of the oxidation of titanium. Akademiya nauk SSSR. Institut metallurgii. Issledovaniye splavov tsvetnykh metallov. no.3. 1962, 116-125. The paper describes an investigation concerned with the initial stages TITLE: of the oxidation of Ti in air and in water vapor and demonstrates that the oxidizing of the oxidation of 11 in air and in water vapor and demonstrates that the oxidizing the growth texture. Floatron-microscopy was the growth texture. SQURCE: medium determines not only the kinetics of the oxidation out also characteristics of the scale formed, such as the type of the growth texture. Electron-microscopy was employed largely. A Ti-O phase diagram is plotted from all available literature. employed largely. A Ti-O phase diagram is plotted from all available literature data including the houndary of the ordering of the gradient actuation. data, including the boundary of the ordering of the a solid solution, more accurate data, including the boundary of the ordering of the w phase; the region of existence information on the composition and boundary of the w phase; the region of existence of a and a process transformation in Ti.O. at 160°C. and a precentation information on the composition and boundary of the ω phase, the region of existence information on the composition and boundary of the ω phase, the region of existence of a and β TiO, the magnetic transformation in Ti2O3 at 160°C, and a presentation of the homologous series of TinO2(n-1) film is practically one single crystal thick of the homologous series of TinO2(n-1) film is practically one single crystal thick sisting of rutile showed that the rutile film is practically one predominantly by a sisting of rutile showed that the rutile film is practically one single crystal thick sisting of rutile showed that the rutile film is practically one single crystal thick sisting of rutile showed that the rutile film is practically one single crystal thick sisting of rutile showed that the rutile film is practically one single crystal thick sisting of rutile showed that the rutile film is practically one single crystal thick sisting of rutile showed that the rutile film is practically one single crystal thick sisting of rutile showed that the rutile film is practically one single crystal thick sisting of rutile showed that the rutile film is practically one single crystal thick sisting of rutile showed that the rutile film is practically one single crystal thick sisting of rutile showed that the rutile film is practically one single crystal thick sisting of rutile showed that the rutile film is practically one single crystal thick sistence for the rutile film is practically one single crystal thick sistence film is practically one sin outward diffusion of Ti ions. This is in agreement with the conclusions of P. Kofstad et al. (Acta Chem Scand no 12 1958 239-280) relative to the kinetics of scale outward diffusion of 11 1008. This is in agreement with the conclusions of F. K. et al. (Acta Chem. Scand., no.12, 1958, 239-280) relative to the kinetics of scale

Card 1/4

On the first stages of the oxidation of titanium.

S/806/62/000/003/010/018

formation at 300-600°. The Kofstad mechanism is briefly summarized and present electron-microscope evidence is adduced to confirm it. With increasing T and oxidation time the formation of more and more sharply bounded rutile crystals was observed, the exterior shape of which is consistent with textbook examples of the habitus of rutile crystals and twins. Disagreement with Kofstad's interpretation of the experimental facts is based on the following reasonings: Kofstad assumed that the determining element in the scale formation is the diffusion of O ions toward the scale-metal discontinuity surface. He oxidized the Ti after rolling, under the assumption that the rolling texture of the Ti would be reflected in the texture of the scale. The appearance of a texture in the scale was then taken to represent a confirmation of the assumption. If that were so, then the oriented texture would follow the texture of the metal best where the scale layer is extremely thin and - in thick scale - where the scale layer adhered most completely to the metal. Neither of these phenomena was observed by Kofstad; the exact opposite was observed in the present investigation (photographs). That rolling texture cannot have much influence on the texture of the scale is also reasonably concluded from the fact that the oxidation occurs in conditions (850°C) in which the Ti specimen is annealed. It is shown that the scale texture cannot be due to recrystallization of the TiO2 and that the only admissible mechanism for scale formation is its formation on its exterior surface and consists of a reaction between the Ti ions diffusing outwardly from the interior

Card 2/4

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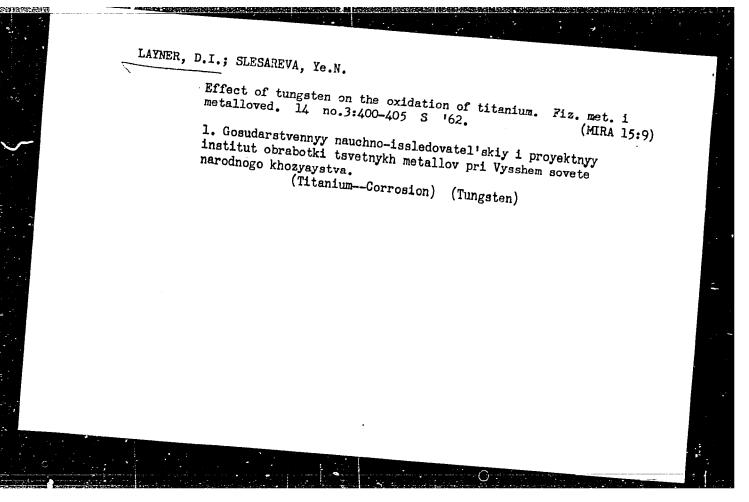
Other investigations have not been able to identify any orienta-On the first stages of the oxidation of titanium. and the scale texture at all, and this is attributed to the preliminary grinding and etching of the Ti-specimen surface which resulted in a disordered microrelief comparable in height with that of the rutile crystals (1 \mu). The rolling-andcomparable in neight with that of the rutile crystals (1 µ). The rolling-and-annealing method used to prepare the specimens in the present investigation yielded a microrelief a full order of magnitude lower and two orders of magnitude less dense (in humps per running mm) than the grind-and-etch method and, hence, interfered less with any terminal tendency of the coals would make according uense (in numps per running mm) than the grind-and-etth method and, hence, interfered less with any textural tendency of the scale growth. The second set of experiments, with oxidation accomplished in water water that the contract of the scale growth. experiments, with oxidation but also a fundamentally afformation of oxidation accomplished in water vapor, showed not only much greater intensity of oxidation but also a fundamentally different type of oxidation Cı process than in air: With increasing T and time the outer scale surface becomes increasingly smooth because the crystals formed in steam oridation. Oχ process man in air. With increasing I and time the outer scale surface becomes increasingly smooth, because the crystals formed in steam oxidation grow so large that their flat surface becomes relatively large with respect to their intercental. is that their flat surface becomes relatively large with respect to their inter-crystal tow boundaries. Then, once the rutile crystals grow beyond 5 \mu, a fundamentally novel hcu: stage occurs, with the rapid growth of thin acicular crystals at separate points of grev stage occurs, with the rapid growth of thin acicular crystals at separate points of the surface (photograph). These needles were found to be near-perfect rutile single-crystals as demonstrated by the evictoric of Kibuchi lines in the microdiffunctional layeı crystals as demonstrated by the existence of Kikuchi lines in the microdiffractional vesti pictures. With increasing T and time the needles grow in length and thickness and The tv occupy an increasing portion of the visual field of the electron microscope. They lamell. occupy an increasing portion of the vibual field of the electron microscope. They become firmly attached to the underlying rutile layers, until the replicas show no in a sei layers: figures, ^{l a}ppear

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Card 4/4

Card 3/4

which 7 are Soviet and - unglish-language).



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18, 17-00

S/136/62/000/011/001/002 E021/E435

AUTHORS: Layner, D.I.

Layner, D.I., Solov'yev, V.Ya., Krupnikova-Perlina, Ye.I., Kachur, Ye.V.

TITLE:

Study of the deformation texture of rolled niobium

PERIODICAL: Tsvetnyye metally, no.11, 1962, 80-85

The main orientations in rolled niobium and the influence of the degree of deformation and the effect of some impurities on the texture of the deformation were studied. Niobium prepared by both the carbon-thermal and the sodium-thermal methods was used. Some cf the niobium was vacuum-sintered at 2300°C in the form of bars 20 \ge 20 \times 600 mm, some was remelted in a vacuum-arc furnace to 70 nm diameter bars and some was remelted in an electron-beam Some of the bars were forged furnace to 80 mm diameter bars. and then cold rolled with intermediate annealing; the total The deformation texture was then compared deformation was 83%. ·for the different starting materials which contained different amounts of impurities $(0_2 - 0.152 \text{ and } 0.083, N_2 - 0.04 \text{ and } 0.1,$ C - 0.04 and 0.07, Si - 0.012 and 0.003 for sodium-and carbonthermal methods respectively). The influence of the melting Card 1/2

s/126/62/013/004/010/022 E111/E435

Layner, D.I., Tsypin, M.I.

Some problems of the formation of scale on metals AUTHORS:

PERIODICAL: Fizika metallov i metallovedeniye, v.13, no.4, 1962,

On the example of titanium the optimum conditions for revealing texture in oxide layers are determined. diffraction patterns indicated texture in the surface layers, probably associated with the surface geometry of their initial The best starting material from surface-preparation aspects for studying scale on sheets is one annealed after cold-In studying texture in scale, identification of orientation with growth or orientation compatibility texture From the observations made it is evident that the surface of the scale is where the main scale component (rutile) is formed, but the interpretation of similar results by P.Kofstad, K.Hauffe and H.Kjöllesdal was incorrect. nossible cause of scale texture is recrystallization of titanium dioxide on heating at up to 850°C for three hours, thus leaving Card 1/3

5/126/62/013/004/010/022 E111/E435

Some problems of the formation ...

growth texture as the cause (for oxidation in air, steam and oxygen); this has not been confirmed by the present authors. Examination of scale obtained under various conditions suggests that the formation of a new layer of scale depends essentially on Published views on these surface and not volume diffusion. The above results relate to questions are contradictory. Next, oxide layers, only some relatively thick scale layers. hundres of angstroms thick, are considered. Microdiffraction investigation (with an electron microscope) has shown that within the boundaries of a grain of the original metal the oxide is precisely orientated, but as this orientation varies from grain to grain the macroscopic orientation of the oxide crystals is random: thus, failure to detect texture by ordinary X-ray and electron diffraction studies of thin films does not indicate absence of connection between oxide orientation and base structure. As scale thickens, orientation differences between adjacent colonies of crystals rather than between oxide crystals affect the orientation within the boundaries of a micro-grain of the original In the case of growth of needle-like rutile crystals, Card 2/3

LAYNER, D.I.; BAY, A.S.

Mechanism of titanium oxidation. Fiz. met. i metalloved. 14 no.2:283-286 Ag '62. (MIRA 15:12)

l. Gosudarstvennyy nauchno-issledovatel skiy i proyektnyy institut obrabotki tsvetnykh metallov.

(Titanium-Metallography) (Oxidation)

5/032/62/028/006/015/025 B101/B138

Layner, D. I., Krupnikova, Ye. I., and Bay, A. S.

AUTHORS:

Electron microscopic determination of the preponderant

TITLE:

orientation of polycrystalline materials

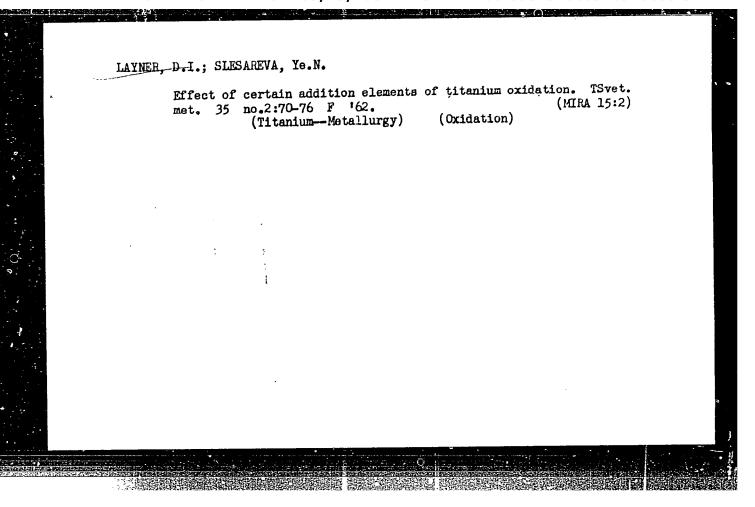
Zavodskaya laboratoriya, v. 28, no. 6, 1962, 703 - 705

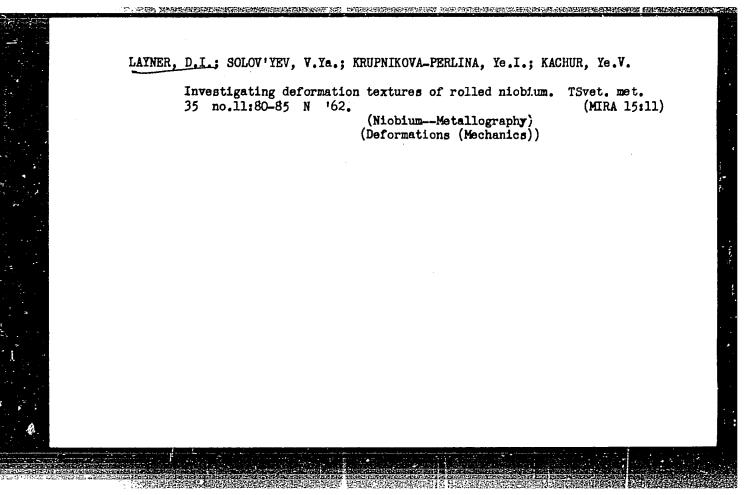
TEXT: A report is given on the determination of the texture of polycrystalline materials by electron microscopic recording of etching figures. Coppernickel 80/20, annealed copper and Cu20, obtained by oxidation of Cu PERIODICAL: at 1020°C and quenching in water, were investigated. Etching of coppernickel was carried out in a mixture of 50% acetic acid and 50% nitric acid, etching of Cu in 50% HNO₃, and of Cu₂O in 10 - 15% HNO₃. The etching

figures were investigated with an 3M-3 (EM-3) electron microscope using carbon replicas. The results agreed well with the data obtained from X-ray recordings of the pole figures. Results: (1) Annealed coppernickel de-

formed by 96% showed two types of etching figures, narrow grooves corresponding to the orientation (110), and lamellas with jagged edges,

Card 1/2





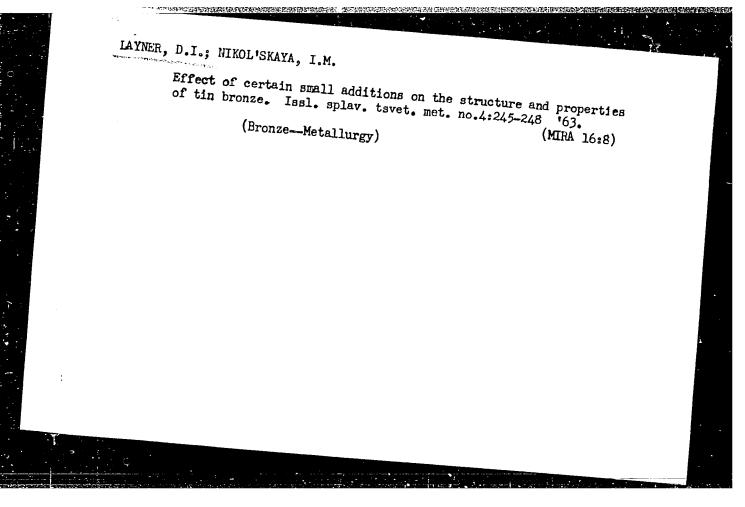
TURETSKAYA, R.A.; GOLUBTSOV, S.A.; TROFIMOVA, I.V.; ANDRIANOV, K.A.;
Prinimali uchastiye: LAYNER, D.I.; SOTNIKOVA, L.I.;

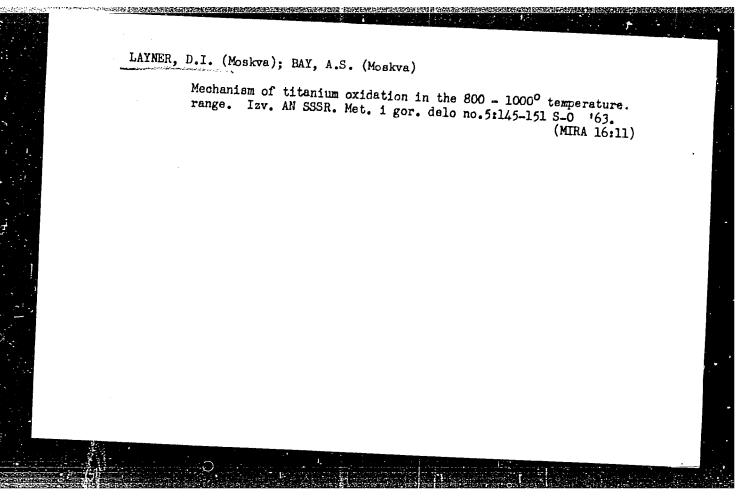
MALYSHEVA, L.A.

Effect of the admixture of some metals on the activity of silicon-copper alloys in the reaction with theyl chloride.
Zhur, prikl, khim. 35 no.7:1496-1502 Jl '62. (MIRA 15:8)

1. Gosudarstvennyy nauchno-issledovatel'skiy i proyektnyy institut obrabotki tsvetnykh metallov (for Layner, Sotnikova, Malysheva).

(Silicon-copper alloys) (Ethane) (Metals)

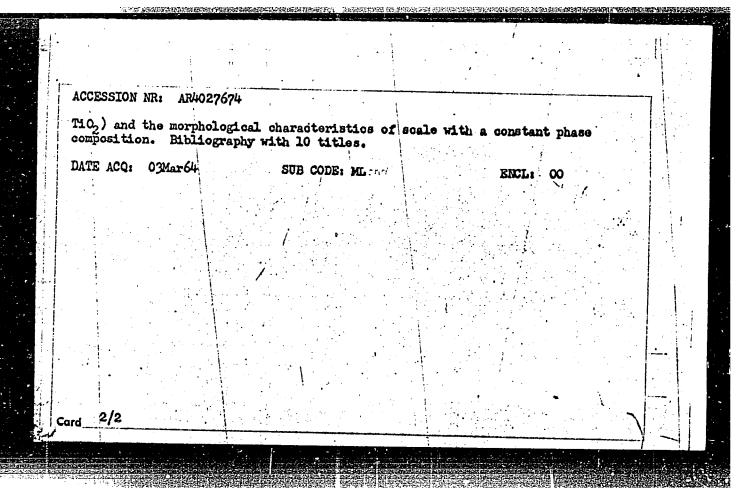




LOBUSEVICH, N.P.; LAYNER, D.I.; TROFIMOVA, I.V.; MALYSHEVA, L.A.;

Reactions of alkyl (aryl) chlorosilane formation by the direct interaction between alkyl (aryl) chlorides and silicon. Report No.5; methyl chloride. of silicon.copper contact masses in reactions with head of the control of t

ACCESSION NR: AR4027674 S/0276/64/000/001/B078/B078 SOURCE: RZh. Tekhnologiya mashinostroyeniya, Abs. 18435 AUTHOR: Layner, D. T.; Bay, A. S.; Tsy*pin, M. I. Some peculiarities of titamium oxidation in various media CITED SOURCE: : Tr. Gos. n.=1. i proyektn. in-ta splavov i obrabotki tsvetn. met., TOPIC TAGS: titanium, titanium oxidation TRANSLATION: The authors present the results of studies on the peculiarities of titanium oxidation in air and water vapor media. It is shown that rutile formed during titanium oxidation in air has a composition which is more closely stoichiometric than that formed in the presence of water vapor or carbon oxides, which in all probability tends to increase the rate of titanium and oxygen ion diffusion. It can be supposed that the acceleration of the oxidation process under these conditions leads to the simultaneous alteration of the composition of scale on the titanium (a deviation from the stoichiometric relationship for Card 1/2



ACCESSION NR: AR4018340

8/0137/64/000/001/1099/1099

SOURCE: RZh. Metallurgiya, Abs. 11645

AUTHOR: Layner, D. I.; Tay+pin, M. I.; Bay, A. S.

TITLE: Electron microscopic study of the scale structure on titanium

CITED SOURCE: Tr. Gos. n.-i. i proyektn. in-ta splavov i obrabotki tsvetn. met., vy*p. 21, 1963, 69-78

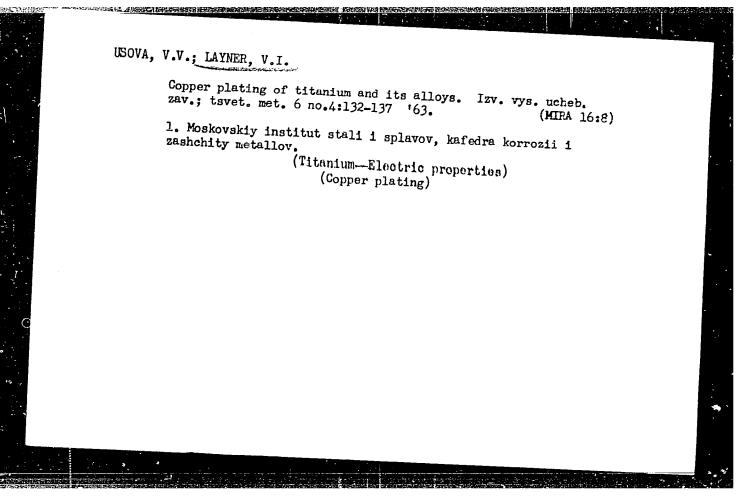
TOPIC TAGS: titanium scale, titanium oxidation

TRANSLATION: Cr-shadowed replicas of Ti scale were studied with an EM-3 electron microscope having a resolving power of ~ 100 Å. When Ti is oxidized in air, the coarse outer crystalline layer does not penetrate into the interior of the scale even under drastic oxidation conditions (900 C, 5 hr). When Ti is oxidized in steem, the columnar layer of scale on Ti, formed by large fused account crystals, makes up the bulk of the scale. L. Petrova

SUB CODE: 1994

ENGL: 00

Cord 1/1



ACCESSION NR: AP3000790 EWP(q)/EWT(m)/BDS AFFTC/ASD AUTHOR: Layner, D. I.; Tsy*pin, M. I.; Bay, A. S. \$/0070/63/008/003/0477/0478 Structural relation between metal and oxide during oxidation SOURCE: Kristallografiya, v. 8, no. 3, 1963, 477-478 TOPIC TAGS: martensite transformation, reciprocal lattice, electron microscope, ABSTRACT: This study was undertaken because of disagreement relative to the existence and nature of interconnection between structures in metals and the coating of secondary products formed by chemical reaction. Previous work by two of the authors (Layner, D. I.; Tsy*pin, M. I. Izv. AN SSSR. Otd. tekhn. n. Metallurof systematic study of these reaction films. This difficulty has now disappeared through application of microdiffraction technique with the aid of an electron microscope to The orientation of individual grains in the polycrystalline metal was ascertained by electron-optical representation of surface relief by means of an escertained by electron-optical representation of surface relief by media of an existence optical and the microdiffraction picture obtained from individual segments of the oxide film was then compared with the orientations of the oxides and the original metal. The authors discovered well-defined martensite structure in their Cara 1/2

L 12788-63 ACCESSION NR: AP3000790

investigation. The experimental data obtained indicate that the oxidation film formed by low-temperature oxidation of titanium in air and water vapor is on the order of 10 sup -6 cm thick and consists of rutile crystals, the orientation of which defines the orientation of original metal grains and is uniform within this grain. Orientations on neighboring grains are unrelated. Orig. art. has: 1

ASSOCIATION: Gosudarstvenny*y nauchno-issladovatel'skiy'i proyektny*y institut splayov i obrabotki tsvetny*kh metallov (State Scientific Research and Planning Institute for Alloys and the Treatment of Nonferrous Metals)

SUBMITTED: 17Mey62

DATE ACQ: 21Jun63

ENCL: 00

SUB CODE: 00

NO REF SOV: 005

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Card 2/2

LAYNER, D.I.; PETRUSEVICH, R.L.; SHIL'SHTEYN, S.Sh.

Theory of a two-crystal spectrometer. Kristalografiia 3 nc.5: 711-714 S-0 '63. (MIRA 16:10)

1. Gosudarstvennyy nauchno-issledovatel'skiy i proyektnyy institut splavov i obrabotki tsvetnykh metallov.

LAYNER, D.I.; BAY, A.S.; TSYPIN, M.I.

Kinetics of oxidation and the structure of scale on titanium. Fiz. met. 1 metalloved. 16 no.2:225-231 Ag *63. (MIRA 16:8)

1. Gosudarstvennyy nauchno-issledovatel'skiy i proyektnyy institut obrabotki tsvetnykh metallov.

(Titanium-Metallography) (Oxidation)

LAYNER, D.I.; TSYPIN, M.I.; BAY, A.S.

Microdiffraction study of the low temperature oxidation of polycrystalline materials. Zav. lab. 29 no.9:1093-1095 '63.

(MIRA 17:1)

1. Gosudarstvennyy nauchno-issledovatel'skiy i proyektnyy institut splavov i obrabotki tsvetnykh metallov.

LAYNER, D.I.; MALYSHEVA, L.A.; YEMEL'YANOV, L.G.; TROFIMOVA, I.V.;

LOBUSEVICH, N.P.; GOLUBTSOV, S.A.

Rate of cooling silicon-copper alloys. TSvet. met. 36 no.8:
76-79 Ag '63.

(MIRA 16:9)

(Nonferrous ingots-Cooling)

L 36297-65 EWT(m)/EWA(d)/EPR/T/EWP(t)/EWP(b)/EWA(c) Parallel Paral

marker. After a three-r was found inside an inte Cu _{l A} lo layer was noted conditions of a cladding	rmediary layer and the	possible to chose the	
ASSOCIATION: None SUBMITTED: 00	ENCL: 00	SUB CODE: MM	
NR REF SOV: 003	OTHER: 002		
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ACCESSION NR: AP4044553

S/0204/64/004/004/0547/0551

AUTHOR: Freydlin, L. Kh., Borunova, N. V., Gvinter, L. I., Layner, D. I., Kagan, N.M.

TITLE: Investigation of the effect of cadmium on the activity and selectivity of nickel-zinc catalysts during hydrogenation of hydrocarbons

SOURCE: Neftekhimiya, v. 4, no. 4, 1964, 547-551

TOPIC TAGS: cadmium, nickel, zinc, nickel zinc catalyst, hydrogenation, catalyst selectivity, hydrocarbon, benzene, styrene, cyclohexene, octene, gas chromatography, catalytic hydrogenation

ABSTRACT: The effect of metallic cadmium on the activity and selectivity of nickel over zinc oxide catalysts during the hydrogenation of hydrocarbons, such as hepten -3(b.p. 95.8-96.1C, $n^{20} = 1.4033$), a mixture of octenes (b.p. 123-125C, $n^{20} = 1.4140$), cyclohexene (b.p. 83C), $n^{20} = 1.4450$), styrene (b.p. 52-53 C/28mm Hg, $n^{20} = 1.5462$; and benzene

(b. p. 80.1C, $n_D^{20} = 1.5017$), was investigated under flow conditions. After cooling to -5C,

Card 1/3

ACCESSION NR: AP4044553

the products were analyzed by gas chromatography. It was found that the relative amounts of cadmium necessary for deactivating the catalyst in the hydrogenation of benzene, cyclohexene and the ethyl bond of styrene were 0.2, 25 and 500% by weight. The probable mechanism of the action of cadmium at different temperatures was studied and discussed. It was established that a variation in the amount of Cd permits the selective hydrogenation of olefins in the presence of benzene or of styrene mited with cyclohexene. The change in the catalytic properties of nickel due to the addition of Cd is due to the change in the composition and crystal structure of the surface layer of the catalyst. Under conditions close to those of the preparation of Ni-ZnO-Cd, cadmium interacts with nickel and forms an intermetallic compound. X-ray analysis and comparison of the interplanar spacings obtained previously showed that the reaction products of mixtures containing up to 70% Cd consist of nickel crystals and D-phase crystals (Cd₁, Ni₁). For products containing only 30% nickel, there was only one line of D-phase with a further increase in the Cd content in the mixture, lines of other intermetallic compounds, apparently with a higher cadmium content (D-phase), appear. On increasing the time of reaction of the catalysts, the loss in Cd increases. New active surface sites on the Ni catalyst are set free and the activity

Card 2/3

ACCESSION NR: AP4044553

increases. Using a catalyst poisoned with 5% Cd the degree of hydrogenation of pentene-3 was 25% after reduction for 10 hours and 90% after 40 hours. Orig. art. has: 4 figures and 2 tables.

ASSOCIATION: Institut organicheskoy khimii im. N. D. Zelinskogo, AN SSSR (Institute of Organic Chemistry, AN SSSR); Gosudarstvenny*ynauchno-issledovatel'skiy institut splavov i obrabotki tsvetny*kh metallov (State Scientific Research Institute for Alloys and Non-Ferrous Metallurgy)

SUBMITTED: 02Jul63

SUB CODE: OC

NO REF SOV: 005

OTHER: 007

Card 3/3

EFR/ENT(m)/EMP(k)/EMP(z)/EMA(c)/EMP(b)/T/EMA(d)/EMP(t) Pf-lu/Ps-lu IJP(c) MJW/JD/HW UR/0126/64/018/001/0145/0145 ACCESSION NR: AP4042811. AUTHOR: Layner, D. I.; Kurakin, A. K. TITLE: Mechanism of the effect of ellicon present in aluminum on the processes of reaction diffusion of iron into aluminum SOURCE: Fizika metallov i metallovedeniye, v. 18, no. 1, 1964, 145-148 TOPIC TAGS: solid state diffusion, iron compound, silicon compound, aluminum compound, iron diffusion, clad iron The purpose of this study was to confirm or refute the formation of the ternary compound Al_xSi_yFe_z. In order to study the effect of silicon on the diffusion of iron into aluminum, an alloy was made from AV000 kluminum and silicon (1.2%). Bimetallic iron-aluminum specimens were prepared by rolling iron and coating it with layers of the Al-Si alloy 2 and 6µ thick. After being vacuum annealed, the specimens were studied by microscopic, x-ray structural, and electron diffraction analysis. The x-ray study showed the absence of a diffusion layer between iron and the aluminum alloy up to 550C. To determine the nature of the reaction diffusion in very thin layers at the interface between the two metals, the authors separated these layers by chemical means (etching), then Card 1/2

ACCESSION NR: AP4042811			2
analyzed them by electron difficition involves the formation temperature rose to just below diffusion layer were observed cover the entire surface of the iron clad with aluminum contachemical compound FeSiAl ₅ . ASSOCIATION: Giprotsvetme SUBMITTED: 07Aug63	v 500C, no visible changes in At 500C, a thin boundary la especimen. The experiments ining 1.5% silicon causes the Orig. art. has: 1 figure and etobrabotka ENCL 00	the phase composition yer of FeSiAl5 was fou show that the annealing formation of the terms	of the and to use of the control of
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L 17534-65 ENT(m)/EPF(c)/ENP(j)/T/ENP(t)/ENP(b) Pc-4/Pr-4 IJF(c)/ASD(')-2/'
ACCESSION NR: AP4044197 S/0079/64/034/008/2706/2708

AUTHORS: Lobusevich, N.P.; Trofimova, I.V.; Golubtsov, S.A.; Andrianov, K.A.; Layner, D.I.; Maly*sheva, L.A.

TITLE: The effect of additions of certain elements to silicon colver alloys on their activity in the reaction with methyl chloride 27

SOURCE: Zhurnal obshchey khimii, v. 34, no. 8, 1964, 2706-2708

TOPIC TACS: salicon copper alloy, methyl chloride reaction, methyl-chlorosilane, synthesis, dimethyldichlorosilane, reaction promoter, reaction inhibitor, phosphorus, sulfur, beryllium, zinc, arsenic

ABSTRACT: The effect of phosphorus, sulfur, beryllium, zinc and arsenic on the overall and the selective activity of Si-Cu alloys in the direct synthesis of methylchlorosilanes was investigated. 0.005-0.008% of F or S and <0.1% of Be lowered the activity of the Si-Cu alloys as determined by the dimethyldichlorosilane yield. 0.05-0.1% As and 0.5-1.5% Zn acted as promoters, increasing the overall and the selective activity of the alloy and lowering the synthesis temperature from 360 to 320C. The nature of the effect of each additive changed depending on the presence of other impurities.

L 17534-65
ACCESSION NR: AP4044197

Thus, P or Be, added to the alloy together with a promoter (Zn), significantly improved the catalyst properties of the Cu-Si alloys, reducing synthesis temperature by 20-40 degrees while increasing the yield of dimethyldichlorosilane to 75%. Orig. art. has: 7 figures and 1 table.

ASSOCIATION: None

SUBMITTED: 03May63

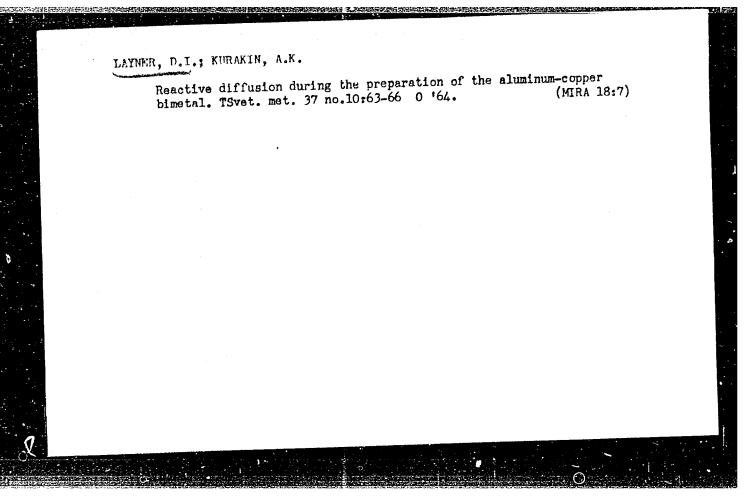
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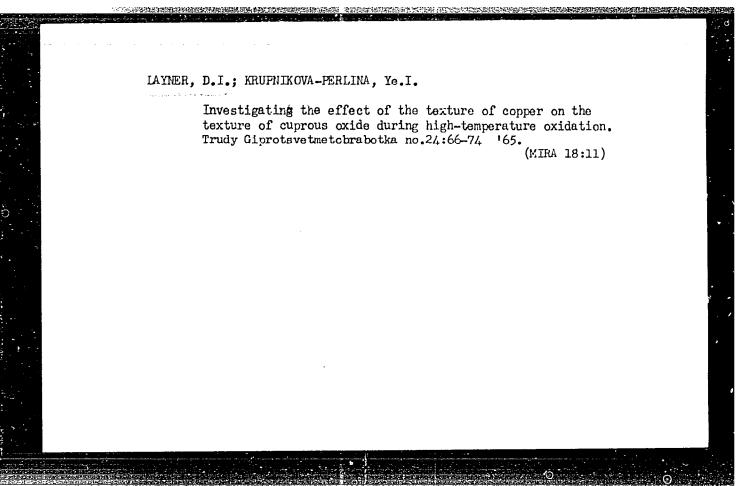
SUB CODE: MM, GC

NR REF SOV: 003

OTHER: 001

Ca. 3 2/2





LAYNER, D.I.; BAY, A.S.

Certain characteristics of the mechanism of titanium and zirconium oxidation. Trudy Giprotsvetmetobrabotka no.24: 93-95 '65.

Growth patterns on the surface of titanium scale.

Ibid.:96-101 (MIRA 18:11)

LAYNER, D.I.; KURAKIN, A.K.

Reactive diffusion of iron in aluminum. Trudy Giprotsvet-metobrabotka no.24:124-130 '65.

Effect of the copper content in aluminum on processes of the reactive diffusion of nickel in aluminum. Ibid.:131-138 (MIRA 18:11)

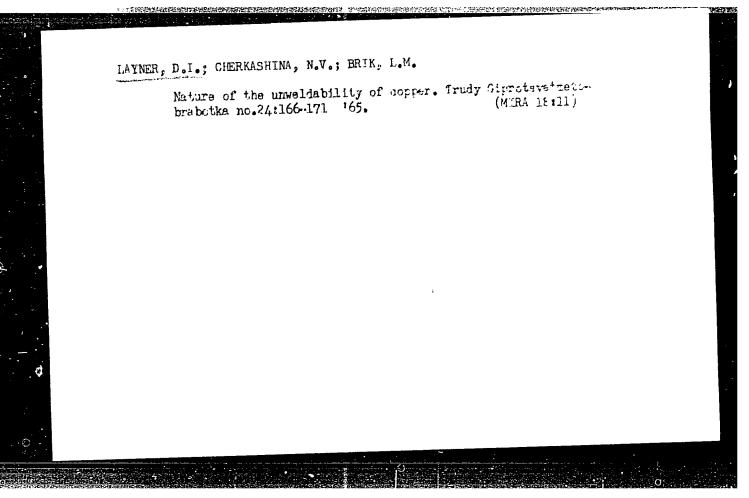
LAYNER, D.I.; SOLOV'YEV, V.Ta.; KHZNETSOVA, M.I.; KRUPHIKOVA-PERLINA, Ye.I.; SLESAREVA, Ye.N.

Studying the exidation of niobium. Trudy dipritsvetestobrabotka no.24:75-85 '65.

(MIRA 18:11)

Mechanism of the electric conductivity of rutile (TiO₂)

and the applicability of the Vagner-Khauffe theory to the oxidation processes of titanium and its alloys. Trudy Giprotsvetmetobrabotka no.24:86-92 165. (MIRA 18:11)



LAYNER, D.I.; OSTROWSKAYA, L.M.; SIMAKOVA, A.S.

Effect of halide impurities on the electric properties of the BigTeg-BigSeg alloy. Trudy diprotevetmetebrabotka no.24: 172-176 '65.

(MIRA 18:11)

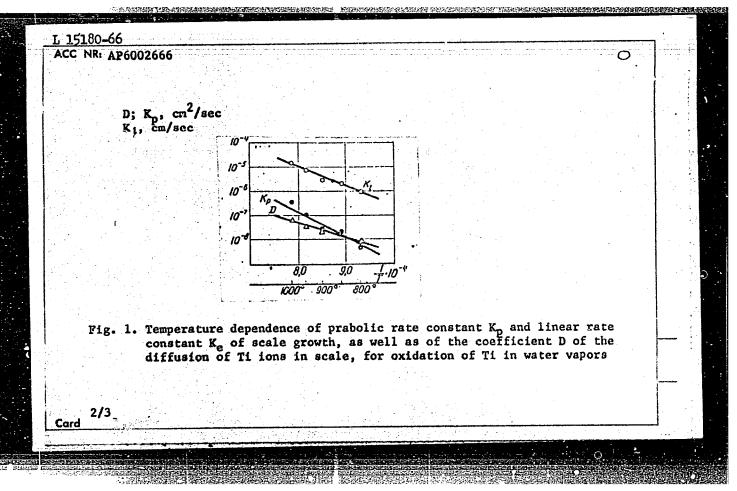
LAYNER, D.I.; MALYSHEVA, L.A.

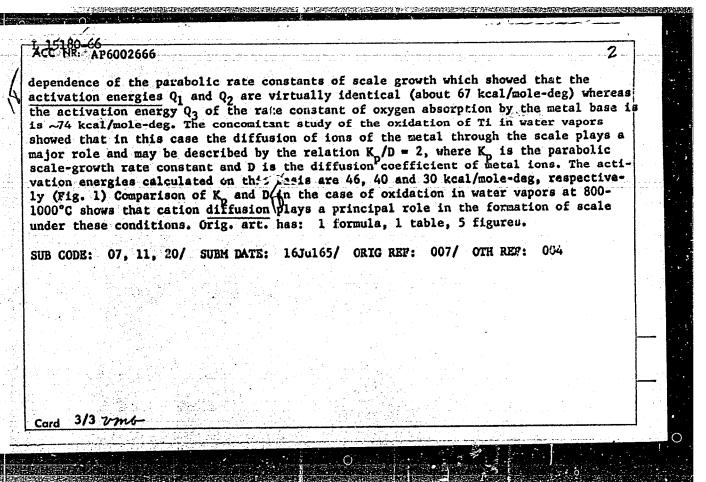
Investigating the effect of small additions of elements of the 5th group of the periodic system on the catalytic activity of silicon-copper alloys. Trudy Giprotsvetmeto-brabotka no.24:146-149 165.

Effect of small additions of bismuth on the Cu₂Si decomposition process in the synthesis of methylchlorosilanes. Ibid.:150-156

Effect of small additions of arsenic on the phase constitution of Cu-Si alloys in the process of synthetizing methylchlorosilanes. Ibid.:157-165 (MIRA 18:11)

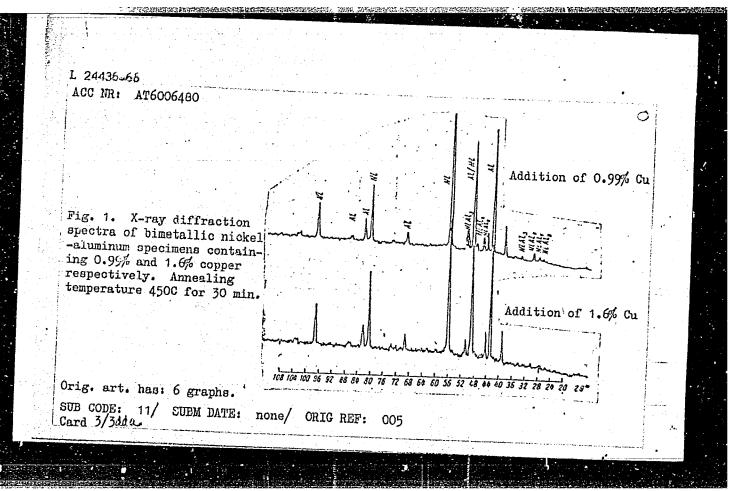
L 15180-66 EMP(e)/EMT(m)/EMA(d)/T/EMP(t)/EMP(k)/EMP(z)/EMP(b) IJP(c) MJW/JD/JW/WB ACC NR: AP6002666 SOURCE CODE: UR/0126/65/020/006/0864/0867	
AUTHOR: Layner, P. I.; Bay, A. S.; Slesareva, Ye. N.; Tsypin, M. I.	
ORG: Giprotsvetmet/brabotka	
TITLE: Certain features of the process of the oxidation of titanium source: Fizika metallov i metallovedeniye, v. 20, no. 6, 1965, 864-867	e de la companya de l
TOPIC TAGS: titanium, metal oxidation, metal scaling, activation energy, cation / VT1-1 titanium ABSTRACT: Some quantitative features of the process of the oxidation of VT1-1 titanium at temperatures above 800°C in an air and water-vapor atmosphere at atmospheric pressure are presented. The published literature specifies the rate constants and activation energy for these regimes only for the case of the oxidation of Ti in 02 and moreover it has been shown that during the oxidation in air of powdered-metal specimens containing 96% Ti the activation energy at temperatures above 800°C differs from the activation energy of oxidation in 02. As for the process of the oxidation of Ti in water vapors at atmospheric pressure, even less is known about it. Accordingly, the authors performed a metallographic study of the oxidation of Ti in air with the object of determining the activation energies of the total absorption of oxygen, scaling, and absorption of oxygen by the metal base, as a function of the temperature	
Card 1/3 UDC: 539.21	
	<u>, 19</u>





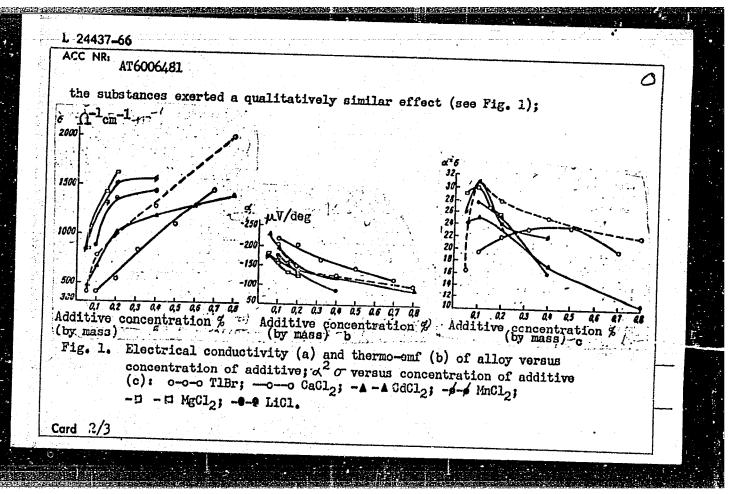
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\$.	24436-66 EWT(m)/EMA(d)/T/EWP(t) IJP(c) JD/HN/JH SOURCE CODE: UR/2680/65/000/024/0131/0138		
	AIN: NRI ALUUUUTUU		•
	AUTHORS: Layner, D. I.; Kurakin, A. K.		
	ORG: State Scientific Research and Design Institute of Alloys and Nonferrous Fitter State Scientific Research and Design Institute of Alloys and Nonferrous Fitter State Scientific Research and Design Institute of Alloys and Nonferrous Fitter State Scientific Research and Design Institute of Alloys and Nonferrous Fitter State Scientific Research and Design Institute of Alloys and Nonferrous Fitter State Scientific Research and Design Institute of Alloys and Nonferrous Fitter State Scientific Research and Design Institute of Alloys and Nonferrous Fitter State Scientific Research and Design Institute of Alloys and Nonferrous Fitter State Scientific Research and Design Institute of Alloys and Nonferrous Fitter State Scientific Research and Design Institute of Alloys and Nonferrous Fitter State Scientific Research and Design Institute of Alloys and Nonferrous Fitter State Scientific Research and Design Institute of Alloys and Nonferrous Fitter State Scientific Research and Design Institute of Alloys and Nonferrous Fitter State Scientific Research and Design Institute of Alloys and Nonferrous Fitter State Scientific Research and Design Institute of Alloys and Nonferrous Fitter State Scientific Research and Design Institute of Alloys and Nonferrous Fitter State Scientific Research and Design Institute of Alloys and Nonferrous Fitter State Scientific Research and Design Institute of Alloys and Nonferrous Fitter State Scientific Research and Design Institute of Alloys and Nonferrous Fitter State Scientific Research and Design Institute of Alloys and Nonferrous Fitter State Scientific Research and Design Institute of Alloys and Nonferrous Fitter State Scientific Research and Design Institute of Alloys and Nonferrous Fitter State Scientific Research and Design Institute Of Alloys A		
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	institut splavov i obradotki tavetajaminim on the diffusion of nickel TITLE: The influence of the copper content in aluminum on the diffusion of nickel		
	into aluminum		
	into aluminum SOURCE: Mosoow. Gosudarstvennyy nauchno-issledovatel skiy i proyektnyy institut SOURCE: Mosoow. Gosudarstvennyy nauchno-issledovatel skiy i proyektnyy institut Trudy, no. 24, 1965. Metallovedeniye i		
	SOURCE: Moscow. Gosudarstvennyy nauchno-issledovatel skly 1 ployedeniye i splavov i obrabotki tsvetnykh metallov. Trudy, no. 24, 1965. Metallovedeniye i splavov i obrabotka tsvetnykh metallov i splavov (Metal science and the treatment of non-obrabotka tsvetnykh metallov), 131-138		
	obrabotka teventykii maa 131 138	4	
	ferrous metals and alloys), 131-138		
	ferrous metals and alloys), 191190 TOPIC TAGS: nickel, aluminum, copper, nickel compound, aluminum plating/ AV000		
	aluminum		
	aluminum ABSTRACT: It was the object of this investigation to extend previously published ABSTRACT: It was the object of this investigation to extend previously published ABSTRACT: It was the object of this investigation to extend previously published ABSTRACT: It was the object of this investigation to extend previously published ABSTRACT: It was the object of this investigation to extend previously published ABSTRACT: It was the object of this investigation to extend previously published ABSTRACT: It was the object of this investigation to extend previously published ABSTRACT: It was the object of this investigation to extend previously published ABSTRACT: It was the object of this investigation to extend previously published	m-	-
	ABSTRACT: It was the object of this investigation to extend previously property of D. I. Layner and A. K. Kurakin (FLM, vyp. 1, 1964, t. 18, str. 145-148) work of D. I. Layner and A. K. Kurakin (FLM, vyp. 1, 1964, t. 18, str. 145-148)	ا اورو حسيد المراد حسيد	
	ABSTRACT: It was the object of this in our of 1, 1964, t. 18, str. 14)-140, work of D. I. Layner and A. K. Kurakin (FMM, vyp. 1, 1964, t. 18, str. 14)-140, work of D. I. Layner and A. K. Kurakin (FMM, vyp. 1, 1964, t. 18, str. 14)-140, and, in particular, to determine whether complex ternary compounds of nickel-coppe and, in particular, to determine whether complex ternary compounds of nickel-coppe and, in particular, to determine whether complex ternary compounds of nickel-coppe and, in particular, to determine whether complex ternary compounds of nickel-coppe and, in particular, to determine whether complex ternary compounds of nickel-coppe and, in particular, to determine whether complex ternary compounds of nickel-coppe and, in particular, to determine whether complex ternary compounds of nickel-coppe and, in particular, to determine whether complex ternary compounds of nickel-coppe and the coppe		1 2
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	are formed at the all also whether the fo					() ()
nickel in	to aluminum. A numb	er of specimens	with aluminum co	pating containin	g from	
10.99 to 4	.08% copper was inverse of 400 to 6500.	estigated. The The specimens we	diffusion was sti re exposed to the	udied in the tem e various tempe	npera- ratures	
for a per	iod of 30 minutes.	After annealing	, the nickel-alur	ninum interface	of the	
specimens	was subjected to minted graphically (se	croscopic and x	ray analysis. '	The experimental mealing of alw	i results	
clad spec	imens with aluminum	coating contain	$_{ m cing}\sim2\%$ Cu lead	is to the format	tion of	-
clad spec	imens with aluminum sting of the ternary	coating contain compound CuzNi	$_{ m Al_6} \sim 2\%$ Cu lead $_{ m Al_6}$ at the nicke	is to the format l-aluminum inter	tion of rface of	
clad specta thin conthe special should re	imens with aluminum ating of the ternary mens. It is suggest tard or prevent the	coating contain compound Cu ₃ Ni ted that the fo diffusion of ni	$aing \sim 2\%$ Cu lead Al ₆ at the nicked expansion of such a ckel into alumina	is to the formation of the second contract of	tion of rface of coating	ale e are Phaeli begar gerben.
clad specta thin conthe special should re	imens with aluminum ating of the ternary mens. It is suggest	coating contain compound Cu ₃ Ni ted that the fo diffusion of ni	$aing \sim 2\%$ Cu lead Al ₆ at the nicked expansion of such a ckel into alumina	is to the formation of the second contract of	tion of rface of coating	ale e ale Paralle par e militaria falla de la companya de la companya de la companya de la companya de la comp
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clad specia thin conthe special should re	imens with aluminum ating of the ternary mens. It is suggest tard or prevent the	coating contain compound Cu ₃ Ni ted that the fo diffusion of ni	$aing \sim 2\%$ Cu lead Al ₆ at the nicked expansion of such a concept into alumination	is to the formation of the second contract of	tion of rface of coating	
clad specia thin conthe special should re	imens with aluminum ating of the ternary mens. It is suggest tard or prevent the	coating contain compound Cu ₃ Ni ted that the fo diffusion of ni	$aing \sim 2\%$ Cu lead Al ₆ at the nicked expansion of such a concept into alumination	is to the formation of the second contract of	tion of rface of coating	



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L 24437-66 EWT(m)/EWP(t)/T/EWP(t) JD/JH ACC 'NR: AT6006481 SQUEER ROOM (ALL)
AUTHORS: Layner, D. I.; Ostrovskaya, L. M.; Simakova, A. S.
Metalworking Mosecy (C. Btl.) Metalworking Mosecy (C. Btl.)
obrabotki tavetnykh metallov)
TITLE: The effect of halide impurities on the electrical properties of Bi2Te3-
SOURCE: Moscow. Gosudarstvennyy nauchno-issledovatel'skiy i proyektnyy institut splavov i obrabotki tsvetnykh metallov. Trudy, no. 24, 1965. Metallovedeniye i ferrous metals and alloys), 172-176
TOPIC TAGS: electric conductivity, thermal emf, ternary alloy, bismuth alloy, halide, electron donor, calcium compound, cadmium compound, lithium compound, manganese compound, thallium compound
ABSTRACT: The effect of chloride salts of calcium, cadmium, magnesium, lithium, and manganess and thallium bromide in concentrations of 0.05—0.8% (by mass) on the properties of a ternary alloy of 80% (mole) Bi ₂ Te ₃ and 20% (mole) Bi ₂ Se ₃ is studied. Melts of 50 g were prepared in sealed quartz ampules at 700C. All of
Card 1/3 Za



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the others were greatly lowered during the first 20—30 days, regardless of storage conditions. Cadmium chloride is recommended as a protective alloy flux. Orig. art. has: 2 graphs and 1 table. SUB CODE: 11/ SUBM DATE: none/ ORIG RFF: 002	ium chloride is recommended as a protective alloy flux. Orig. hs and 1 table.	conditions. Cadmium chloride is recommended as a protective alloy flux. Orig. art. has: 2 graphs and 1 table.
art. has: 2 graphs and 1 table.	hs and 1 table.	art. has: 2 graphs and 1 table.
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1	ACC NR: AT6006476 SOURCE CODE: UR/2680/65/000/024/0093/0095		eru i
•	ATTIMUCES, Tormon D. T., Borr A. C.		11 31
	AUTHORS: Layner, D. I.; Bay, A. S.		
,, .	ORG: State Scientific Research and Design Institute of Alloys and Nonferrous		
	Metalworking, Moscow (Gosudarstvennyy nauchno-issledovatel skiy i proyektnyy institut splavov i obrabotki tsvetnykh metallov)	V	
	TITLE: Certain peculiarities of the oxidation mechanism of the oxidation of titanium and zinconium		
	γ		
	SOURCE: Moscow. Gosudarstvennyy nauchno-issledovatel'skiy i proyektnyy institut		
	splayov i obrabotki tsvetnykh metallov. Trudy, no. 24, 1965. Metallovedeniye i obrabotka tsvetnykh metallov i splayov (Metal science and the treatment of nonferrous		
	metals and alloys), 93-95		
. :	Til virgillation		
	TOPIC TAGS: zirconium, zirconium oxide, electron microscopy, titanium, voridation		
5	ABSTRACT: The object of the investigation was to extend earlier published work on		
	titanium by D. I. Layner, A. S. Bay, and M. I. Tsypin, (FMM, vyp. 2, 1963, t. 16),		
	particularly the study of the microstructure of zirconium after it was exposed to air and water vapor at 10000 for periods of 30 and 120 minutes respectively and to compare		
	the latter with the microstructure of titanium subjected to the same experimental		1
-	conditions (see reference above). The microstructure was studied by means of electron microscopy. Several electron microscope pictures of the zirconium specimens are		
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	Card 1/2	-	

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SUB	CODE:	11/	SUBM	DATE:	none/	ORIG	REF:	002/	OTH	REF:	002		*.			
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1 24434-66 ENT(17 ACC NRI AT6006477 ENT(m)/T/EHP(t) IJP(c) JD/MB SOURCE CODE: UR/2680/65/000/024/0096/0101 AUTHORS: Leyner, D. I.; Bay, A. S. 36 ORG: State Scientific Research and Design Institute of Alloys and Nonferrous 13H Metalworking, Moscow (Gosudarstvennyy nauchno-issledovatel'skiy i proyektnyy institut splavov i obrabotki tsvetnykh metallov) TITLE: Growth forms on the surface of titanium sinter SOURCE: Moscow. Gosudarstvennyy nauchno-issledovatel skiy i proyektnyy institut splavov i obrabotki tsvetnykh metallov. Trudy, no. 24, 1965. Metallovedeniye i obrabotka tsvetnykh metallov i splavov (Metal science and the treatment of nonferrous metals and alloys), 96-101 TOPIC TAGS: titanium, titanium oxide, electron microscopy, oxidation ABSTRACT: The object of this investigation was to extend earlier published work of D. I. Layner, A. S. Bay, and M. I. Tsypin (FMM, vyp. 2, 1963, t. 16, str. 225-231) and, in particular, to investigate the surface structure of titanium after it was oxidized by water vapor at 9000, in air at 850 and 10000, and in oxygen at 12000. investigation was conducted by opical and electron microscopy. Electronmicroscope pictures of the various specimens are presented. It was found that the microstructure of the surface changes with the change in the nature of the oxidizing medium. It is suggested that the observed crystalline shapes arise from a migration of litanium ions. Card 1/2